

Electrocatalytic properties of polyaniline–TiO₂ nanocomposites

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Abstract An efficient methodology of polyaniline–TiO₂ nanocomposite (PAniTNC) was developed through various proportions of ceramic added by chemical polymerization at room temperature. The chemical polymerization reactions were carried out using potassium persulphate as an oxidizing agent. The particle size of PAniTNC was found in the range of 8–15 nm as analysed by transmission electron microscopy. The Ti–O characteristic stretching bands at 630 and 558 cm^{−1}, indicating the presence of TiO₂ in polyaniline matrix was confirmed by fourier transform infrared spectroscopy. The characterization of the nanocomposite crystalline shape was carried out using X-ray diffraction method. Thermogravimetric and differential scanning calorimetry analyses indicate high thermal stability. SEM analysis showed mixed granular nature of the polymer–ceramic nanocomposites. EDX analysis shows the presence of Ti, C and N in composites. Cyclic voltammetric studies exhibit good adherent behaviour on the electrode surface at pH 1.0. It reveals the presence of oxidation peak at 0.417 V and reduction peak at 0.285 V. It has been observed that PAniTNC-modified GCE has good oxygen reduction ability, and also enhances methanol reduction.

Keywords Polyaniline–TiO₂ nanocomposites · Voltammetry · TGA/DTA · SEM · TEM · XRD

Introduction

Several approaches are developed in the recent years for full substitution of platinum in oxygen reduction reaction (ORR) electrocatalysts [1]. It is known that conducting polymers (CPs) such as polyaniline and polypyrrole can directly demonstrate electrocatalytic properties in ORR [2, 3], which is attractive for the development of ORR electrocatalysts on their basis. To enhance catalytic efficiency of CPs in ORR, it is suggested [4, 5] to use Keggin-type heteropoly acids (HPAs) for molecular dispersion of catalytically active particles as specific polymer dopants in the CP matrix.

It was found earlier [6, 7] that three-component nanocomposites are based on CPs (PAni or PPy), HPA and V₂O₅–CP HPA/V₂O₅, and their bifunctional analogues. Nanosize platinum containing CP HPA/V₂O₅/Pt single material and the properties of both CP HPA/V₂O₅ and CP/Pt composites show high electrocatalytic activity in ORR which is due to nanostructuring and interaction between the components. With account for this, it was of interest to elucidate the possibility of using other nanosize transition metal oxides for development of hybrid CP-based nanocomposites capable of manifesting electrocatalytic properties in ORR. Titanium dioxide featuring catalytic activity in a number of chemical, electrochemical and photochemical redox processes attracted our attention which eventually led to the development of metal oxide component with CP-based nanocomposites. In particular [8–10], various TiO₂ forms can demonstrate electrocatalytic properties in ORR in acidic electrolytes and the prevailing process herewith, according to the two-electron oxygen reduction with the formation of hydrogen peroxide. The CP dopant was chosen to be 12-phosphomolybdic acid, H₃PMo₁₂O₄₀ (PMA) that features the highest

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electrocatalytic activity in ORR among the unsubstituted Keggin- or Dawson-type HPAs [11].

Importantly, PANi-inorganic oxide nanocomposites are effective electrocatalysts for many reactions and have been employed in sensor applications [12, 13]. PANi/WO₃ composites have been prepared and used as humidity sensor [14]. The heterogeneous branched core-shell SnO₂-PANi nanorod arrays have been designed and fabricated by an efficient and facile hydrothermal treatment followed by electrodeposition [15]. Phosphomolybdic acid-doped PANi/V₂O₅ composite has been proved to be an effective electrocatalyst in oxygen reduction in weakly acidic solution. Metal oxides such as MoO₃ and VO₂ offered electrocatalytic properties when incorporated into PANi matrix [16–18]. Electroactivity of the NiO(*x*)/POT(*y*) (*x* = amount of NiO-NPs (mg), *y* = concentration of OT (mM)) films showed dependence on *x* and *y*. Efficient electrocatalysts could be obtained by tuning of *x* and *y* was reported by Komathi et al. [19]. Gang Wu et al. reported that the nitrogen-doped TiO₂-supported PANi-Fe catalyst exhibits much improved catalytic activity for oxygen reduction when compared to the traditional carbon-supported materials [20].

Thus, the aim of this work was to study the regularities of electrochemical and electrocatalytic behaviour in the ORR of the hybrid nanocomposites based on CP and nanosize ceramic. Their methanol reduction in different atmospheric conditions is also studied.

Methods

UV–Vis spectra of the samples were recorded on a JASCO V530 UV–Vis spectrophotometer. Fourier transform infrared spectroscopy (FTIR) (Model: SHIMADZU) of polyaniline–TiO₂ nanocomposites (PANiTNCs) was studied in the frequency range of 400–4000 cm^{−1}. X-ray diffraction (XRD) was carried out with the Bruker AXS D8 Advance using CuK_α radiation source energized at 40 kV ($\lambda = 0.154$ nm). Samples were scanned at a rate of 29.5° per sec where 2θ ranges from 1° to 80°. Morphological study of the PANiTNCs was carried out using scanning electron microscopy (SEM Model: JEOL JSM 6360) operated at 25 kV. The structure of the PANiTNC was confirmed using transmission electron microscopy (TEM) (PHILIPS Model CM200) operated at 20–200 kV with resolution of 2–4 Å. Differential scanning calorimetry (DSC) studies were carried out using METTLER Toledo DSC operated at −150 to 700 °C. Thermogravimetric analysis (TGA) studies were carried out using Perkin Elmer Thermal Analysis in which operating temperature ranges between 40.00 and 960.00 °C at a rate of 20.00 °C/min. Cyclic voltammetry, chronoamperometry and chronocoulometry studies were carried out in the presence of

nitrogen and oxygen atmosphere, and impedance spectroscopy was also experimented using CH Instrument (Model 650C) Electrochemical workstation.

Results and discussion

XRD studies

X-ray diffraction patterns were recorded for the prepared samples of PANiTNCs and a comparison was made between them as shown in Fig. 1a–f. The XRD pattern of PANi shows a broad peak at 2θ whose resultant value indicates 25.30° which corresponds to the (110) plane of PANi. It reveals that PANi is amorphous in nature which may be assigned to the scattering of PANi chains at inter-planar spacing. When K₂S₂O₈ is added to the reaction system, it is observed that polymerization proceeds initially on the surface of TiO₂ nanoparticles due to the restrictive effect of the surface. Thereafter, polyaniline encapsulates the crystalline behaviour of TiO₂ thus hampering its crystalline behaviour. Therefore, the degree of crystallinity of polyaniline decreases, and the diffraction peaks emerge with TiO₂ peaks and hence cannot be distinguished. By comparing the XRD patterns of TiO₂-PANi nanocomposite with that of PANi, it was found that the prominent peaks corresponding to $2\theta = 25.61^\circ$, 38.12° , 47.53° and 54.03° are due to (110), (101), (111) and (211) crystal planes of

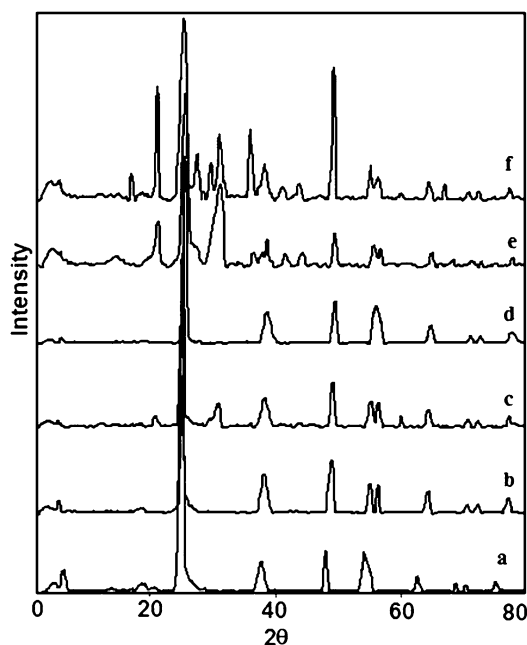


Fig. 1 X-ray diffractogram of PANiTNCs (a PANiTNC1, b PANiTNC2, c PANiTNC3, d PANiTNC4, e PANiTNC5 and f PANiTNC6)



anatase TiO_2 which indicate the presence of TiO_2 in PANi [JCPDS file no. 21-1276]. It is observed that TiO_2 has retained its structure even though it is dispersed in PANi after polymerization reaction [21]. The very sharp intensity peaks obtained for PANiTNC5–PANiTNC6 are not so in the case of PANiTNC1–PANiTNC4 as the peaks are suppressed due to the conducting emeraldine base of PANi.

SEM, EDX and TEM studies

Figure 2a–f shows the SEM images of PANiTNC1–PANiTNC6. It can be seen that the morphology of composites for low contents of TiO_2 i.e. 0.25, 0.50, 0.75 and 1.0 g of TiO_2 is much similar to that of polyaniline. Especially, PANiTNC1 (0.25 g) shows some fibrous microstructure due to the very low composition of TiO_2 nanoparticles. The change in surface morphology has been observed with the increasing content of TiO_2 (0.25–1.50 g) in PANi. The encapsulated TiO_2 particles in the PANi matrix have changed from sponge-like structure (PANiTNC1) to granular flower (PANiTNC6) structure. As the composition of TiO_2 increases, agglomeration and uneven distribution of TiO_2 particles can be seen in the PANi matrix. The diffraction pattern from the edge of particles of PANiTNC1–PANiTNC6 suggests that the TiO_2 nanoparticles are deposited on the surface of PANi and show typical anatase phase, and this agrees well with the X-ray diffraction result of TiO_2 nanoparticle sample [22]. The EDX analyses of the PANiTNCs are also performed in order to confirm the incorporation of the TiO_2 -nanoparticles in the PANi matrix. Intensity of the peak is found to increase gradually while increasing the ceramic amount.

TEM images of the nanocomposite are presented in Fig. 3. In Fig. 3a, aggregation with spherical structures is found obviously and the spherical aggregates are composed of dark spots. As shown in Fig. 3a, the dark spots symbolising TiO_2 nanoparticle are encapsulated with PANi matrix for PANiTNC1 and are found to be larger than the particle size obtained from XRD results. The particle size of ceramic nanomaterial is observed and ranges between 29.6 and 123.6 nm as shown in Fig. 3a. Therefore, it is considered that the aggregates of TiO_2 nanoparticles exist within the PANi matrix. Figure 3b shows the electron diffraction pattern of PANiTNC1.

DSC studies

Differential scanning calorimetry (DSC) of PANiTNC1–PANiTNC6 samples is shown in Fig. 4a–f and Table 1. The first peak appeared in the temperature range of 67–112 °C which may be inferred as the release of moisture. Second peak appeared in the range of 121–212 °C. This may be attributed to the degradation of polymer functionalities.

The third peak appeared in the range of 210–262 °C which may be assigned to the release of loosely bounded polymer molecule in nanocomposites. The fourth peak value is obtained above 300 °C in PANiTNC4–PANiTNC6 due to nanocomposites. These thermal data indicate the softening of polyaniline on addition of TiO_2 thus enhancing their thermoplasticity and thermoprocessing. These results prove the greater thermal stability of the nanocomposite due to the attractive coulombic interaction between the positive group of PANi layer and negatively charged surface of the TiO_2 layer. PANiTNC1 exhibits only two weight losses which is indicative of the low amount of TiO_2 and hence is stable.

TGA and DTA studies

The TGA of PANiTNCs samples is shown in Fig. 5a–f and Table 2. The first step (85–98 °C) may be assigned to the loss of moisture. In the second step (299–322 °C), elimination of the polymer is expected. The result agrees with the evaporation of moisture which is trapped inside the polymer or bound to the polymer backbone as evidenced by the degradation stage of TGA curve [23]. The third weight loss starting from ~760 to 832 °C could be correlated to the thermal decomposition of the nanocomposite [24]. The PANiTNC1–PANiTNC6 matrix shows residual weights around 20 % at 916 °C. Figure 5a (PANiTNC1) shows that it was stable up to 832 °C, when compared to PANiTNC2–PANiTNC6 as well as PANi. The temperatures get gradually decreased while increasing the ceramic content in the PANi matrix. Thus, it could be concluded that PANiTNC1 is thermally stable.

FTIR studies

The important characteristic peaks observed in FTIR spectra of PANiTNCs are shown in Table 3 and Fig. 6a–f. The origins of the vibration bands are as follows: The stretching frequency at 3431–3456 cm^{-1} is due to the NH of aromatic amine. Those from 2852 to 2923 cm^{-1} are due to aromatic CH stretching vibration, and CH out-of-plane bending vibration was observed at 505 cm^{-1} . The CH out-of-plane bending mode has been used as a key to identify the type of substituted benzene. The bands of PANi at 1558, 1577–1581 and 1481–1485 cm^{-1} are attributed to C=N and C=C stretching modes of vibration for the quinonoid and benzenoid units of polyaniline. The peaks at 1298–1301 and 1242–1245 cm^{-1} are assigned to the C–N stretching of benzenoid ring. The peak at 1242 cm^{-1} is characteristic of the conducting protonated form of polyaniline. The bands in the region 1114–1126 cm^{-1} are due to in-plane bending vibrations of C–H mode of N=Q=N, Q=N + H=B and B–N + H–B, which is formed



Fig. 2 SEM and EDX of PAniTNCs (**a** PAniTNC1, **b** PAniTNC2, **c** PAniTNC3, **d** PAniTNC4, **e** PAniTNC5 and **f** PAniTNC6)

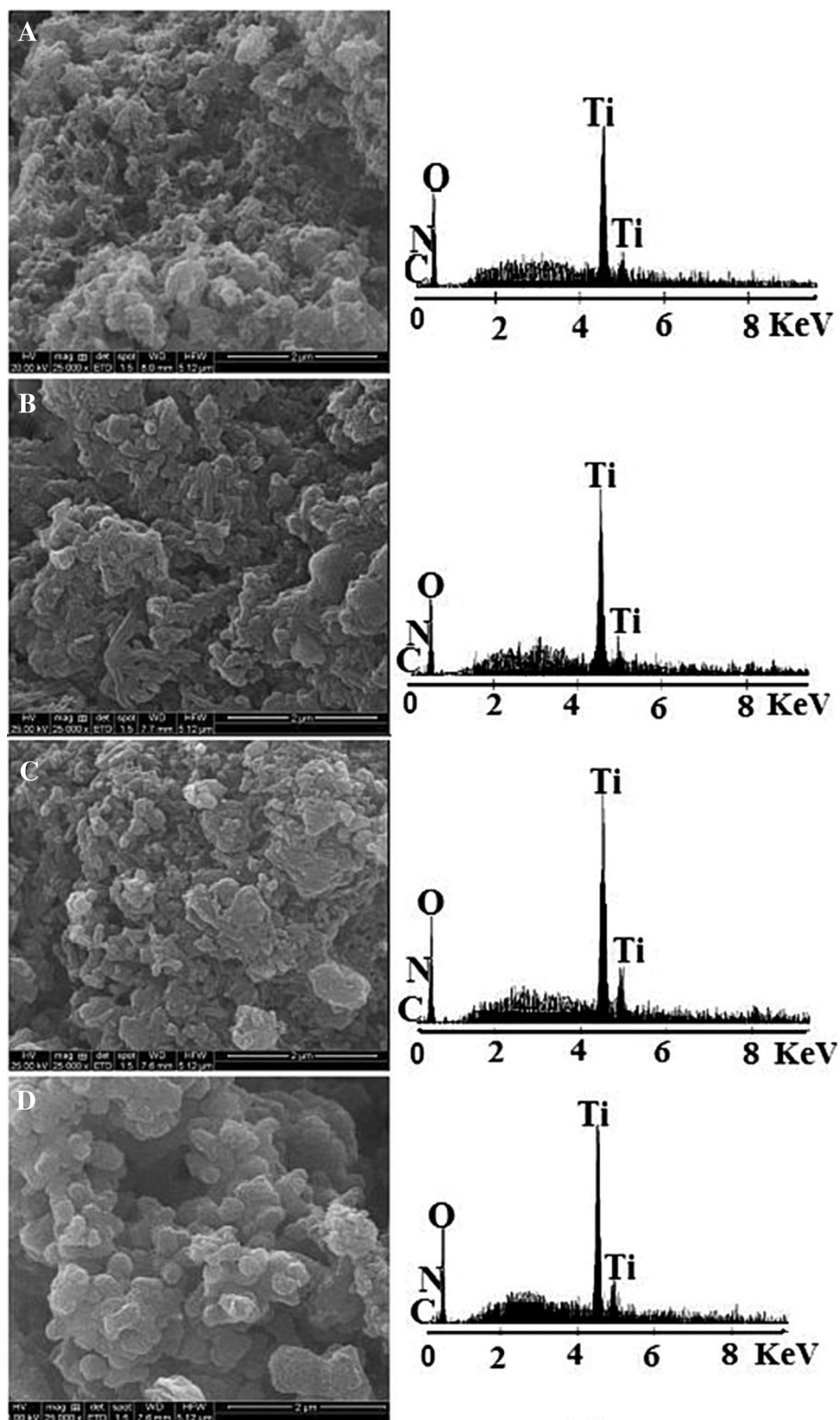
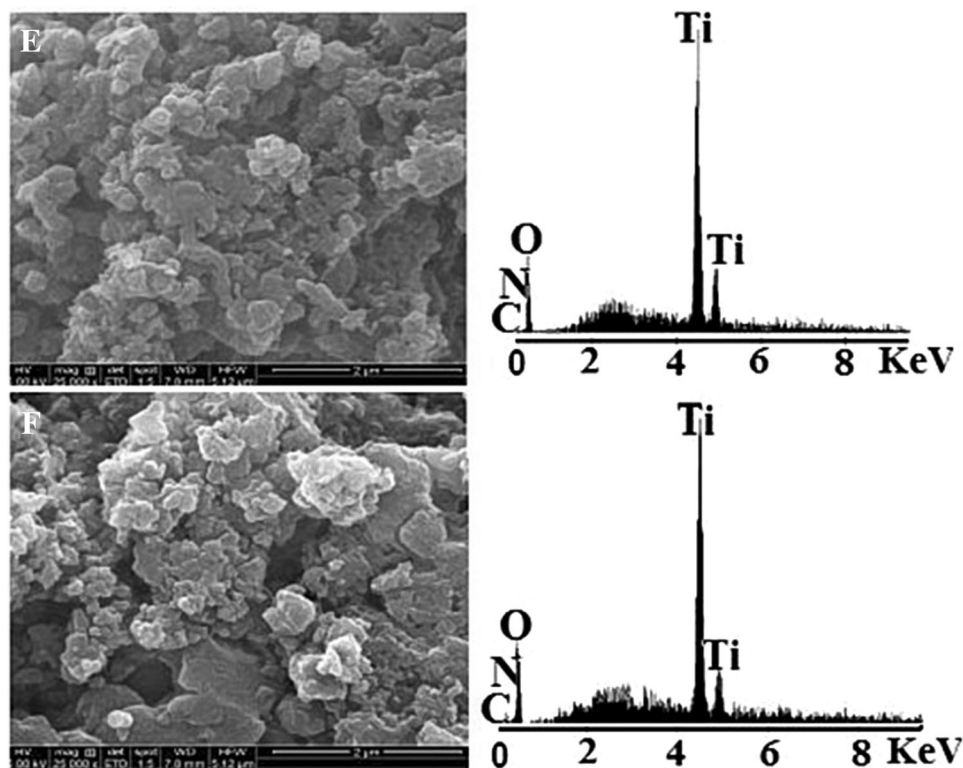


Fig. 2 continued



during protonation [25]. The band at $798\text{--}804\text{ cm}^{-1}$ originates from the out-of-plane C–H bending vibration. The low-wavenumber region exhibits a strong vibration around 725 cm^{-1} which corresponds to the anti-symmetric Ti–O–Ti. When compared with PANi, certain changes were observed in intensity and wavelength shifts due to changes in π electron delocalisation in the aromatic rings; the shift towards higher wavenumber indicates an increase in the conjugation length as a result of doping with an electron-rich dopant. Both the amine and imine groups of PANi are affected by dopant, thus changing the wave number of both C–C and C=N stretching. These red-shifted peaks may also be due to PANi transition from the emeraldine base (EB) to the emeraldine salt (ES) during the doping process or morphological changes [26]. These results confirm that structural and morphological changes occurred in PANi. IR spectral peak value confirms the formation of PANi–titania nanocomposites.

UV–Vis studies

Prepared PANi–TiO₂ nanocomposites were studied by UV–Vis absorption spectroscopy with λ ranging from 1100 to 200 nm. The absorption spectra thus obtained incorporating various ratios of TiO₂ used in PANiTNCs are shown in

Fig. 7a–f. The electronic spectra of PANiTNCs exhibited bands at ~ 272 and $\sim 368\text{ nm}$ which are attributed to $\pi\text{--}\pi^*$ transitions in the benzenoid structure [27]. The peak at about 610 nm has been shifted to lower wavelengths of $\sim 558\text{ nm}$ which may be due to the $n\text{--}\pi^*$ transitions of quinone–imine groups [28]. These results showed that PANi was completely converted from emeraldine salt to the emeraldine base form by the deprotonation of PANi with NH₄OH [29].

The intensity of peaks alters with different concentrations of ceramic content. Figure 7a indicates that insertion of nano-TiO₂ particles led to doping of the conducting polyaniline, and hence led to an interaction at the interface of polyaniline and nano-TiO₂ particles [30, 31]. It has been noticed that these peaks PANiTNC1 shift towards lower intensity when there is a change in ceramic amount (0.5, 0.75, 1.0, 1.25 and 1.5 g). This may be due to interchain species, which play an important role in the process of conjugated polymers. The strong interactions between the PANi chains create conjugation defects leading to a compact coil conformation. As the TiO₂ content is increased (Fig. 7b–f), a free carrier tail is observed in the visible region, which is consistent with the delocalization of polarons [25] promoted by extended chain conformation of PANi.



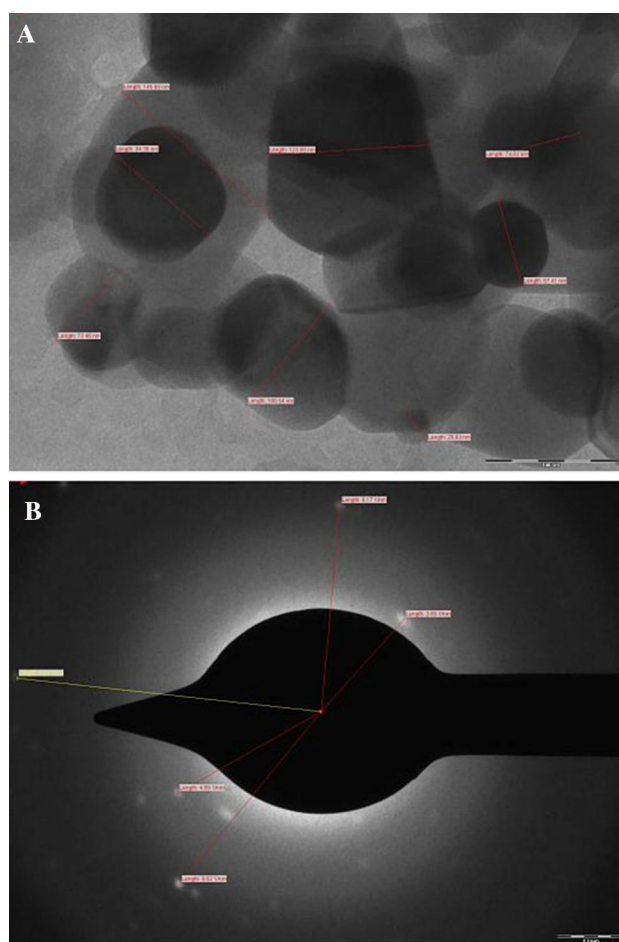


Fig. 3 **a** Transmission electron micrographs of PANiTNC1 and **b** electron diffraction pattern of PANiTNC1

Electrochemical characterization

Cyclic voltammetry

PANiTNC1-modified glassy carbon electrodes were employed in the present investigation. It was dissolved in dimethyl sulfoxide (DMSO) to form polymer–ceramic nanocomposite solution. PANiTNC1 films were produced by casting the solutions (1 drop) onto a clean and pre-treated 0.0314 cm^2 GCE and then allowing the solvent to evaporate. They were consequently rinsed with water and then transferred to an electrochemical cell for experimental purpose. A three electrode cell with a saturated calomel reference electrode (SCE), a platinum wire counter electrode and PANiTNC1-modified glassy carbon working electrode was employed. All electrochemical experiments were carried out at a thermostatic temperature of $25.0 \pm 2^\circ \text{C}$.

Cyclic voltammetric studies of PANiTNC1-modified GCE were performed for different pH, different concentrations and different atmospheric (N_2 and O_2 gases) conditions. The CVs were obtained in $1.0 \text{ M H}_2\text{SO}_4$ electrolyte by casting the nanocomposite on GC working electrode and scanned between -1.2 and 1.2 V at scan rates from 50 to 500 mV/s .

The effect of different concentrations of PANiTNCs on the voltammetric response was studied as shown in Fig. 8a. The high peak current appeared for PANiTNC1 when compared with other PANiTNCs. Thus, the current of PANiTNC1-modified GCE increases with the addition of TiO_2 (0.25 g TiO_2) and decreases with a further increase in TiO_2 . This result clearly shows that the increases the current density with decreasing the amounts of TiO_2 , because conducting polymer matrix is replaced gradually by ceramic. Thus PANiTNC1 gives better voltammetric response and which was chosen for subsequent studies.

Influence of pH

The effect of pH on voltammetric response of PANiTNC1 was studied on modified GCE in the pH range of 1.0 – 13.0 , using pH buffer solutions adjusted to the desired condition which is illustrated in Fig. 8b. Background current or capacitive current is generally pH dependent. When the pH of the solution is greater than seven, PANiTNC1 becomes non-electroactive. The indication of lesser electroactivity is due to the failure of the formation of the emeraldine salt (ES) with titania nanocomposites. The surface of TiO_2 particles has zero charge at pH 7 . This could be made to acquire positive charge in acidic condition by the addition of H_2SO_4 (pH 1 – 3). Thus, the Cl^- ions get adsorbed on the positively charged TiO_2 particles which would work as a charge compensator for positively charged PANi chain in the formation of PANi– TiO_2 nanocomposites [32]. The optimum pH range for good sensitivity response appears to be 1.0 .

Effect of different atmospheres

The voltammograms for PANiTNC1-modified GCE in $1.0 \text{ M H}_2\text{SO}_4$ were obtained in normal as well as N_2 - and O_2 -saturated buffer. These voltammograms are demonstrated in Fig. 8c. Cyclic voltammograms depict a well-defined pair of oxidation–reduction observed on modified GCE at 50 mV/s .

The cyclic voltammogram of PANiTNC1-modified GCE presented in Fig. 8c shows an oxidation peak at $+565.9 \text{ mV}$ and a broad reduction peak which started from $+683.7$ to $+326.7 \text{ mV}$ under normal condition. PANi– TiO_2



Fig. 4 DSC of PAniTNCs (a PAniTNC1, b PAniTNC2, c PAniTNC3, d PAniTNC4, e PAniTNC5 and f PAniTNC6)

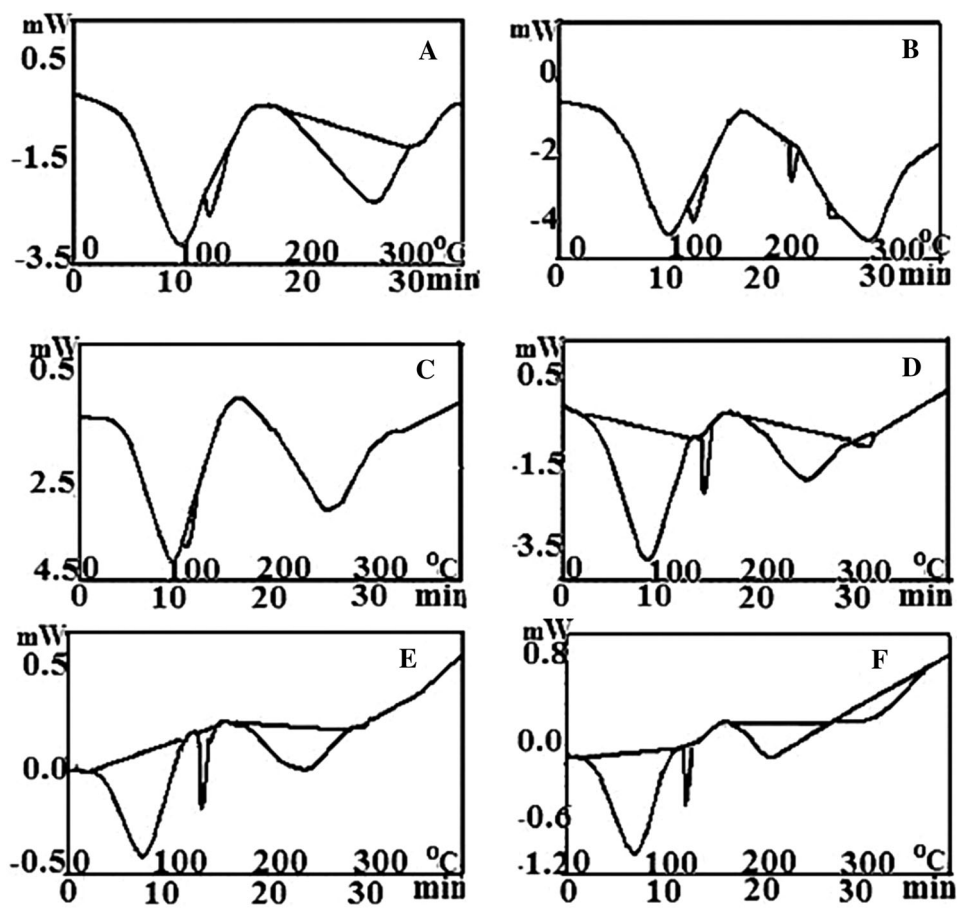


Table 1 DSC values of PAniTNC1–PAniTNC6

PAniTNCs	First peak (°C)	Second peak (°C)	Third peak (°C)	Fourth peak (°C)
PAniTNC1	112.89	–	262.4	–
PAniTNC2	112.5	206.3	262.29	–
PAniTNC3	109.85	190.3	251.1	–
PAniTNC4	84.87	143.7	249.73	318.26
PAniTNC5	71.3	131.8	235.17	302.0
PAniTNC6	67.96	121.3	210.8	322.68

shows better electrochemical catalytic activity. Under N_2 -saturated buffer, it shows an oxidation peak at +469.3 mV and a broad reduction peak from +730.1 to +290.9 mV. At O_2 -saturated buffer, two oxidations at +330.1, +672.8 mV and reduction at +558.7, +162.4 mV are observed. The presence of an additional peak could be related to the degradation product (quinine/hydroquinone couple) [33]. PAniTNC1-modified GCE shows a larger enhancement in the anodic and cathodic current peak and electro-oxidation

peak shifted towards lower potential (~ 130 mV) and the electro-reduction peak potential shifted towards the negative side (~ 180 mV), when compared with N_2 -saturated buffer. The change in conductivity is accounted for the interaction between O_2 and PAni which loses and gains electrons in valence band in the presence of TiO_2 [34]. The result clearly shows that PAniTNC1 has electrocatalytic redox behaviour enhanced under O_2 -saturated buffer.



Fig. 5 TGA/DTA curves of PANiTNCs (a PANiTNC1, b PANiTNC2, c PANiTNC3, d PANiTNC4, e PANiTNC5 and f PANiTNC6)

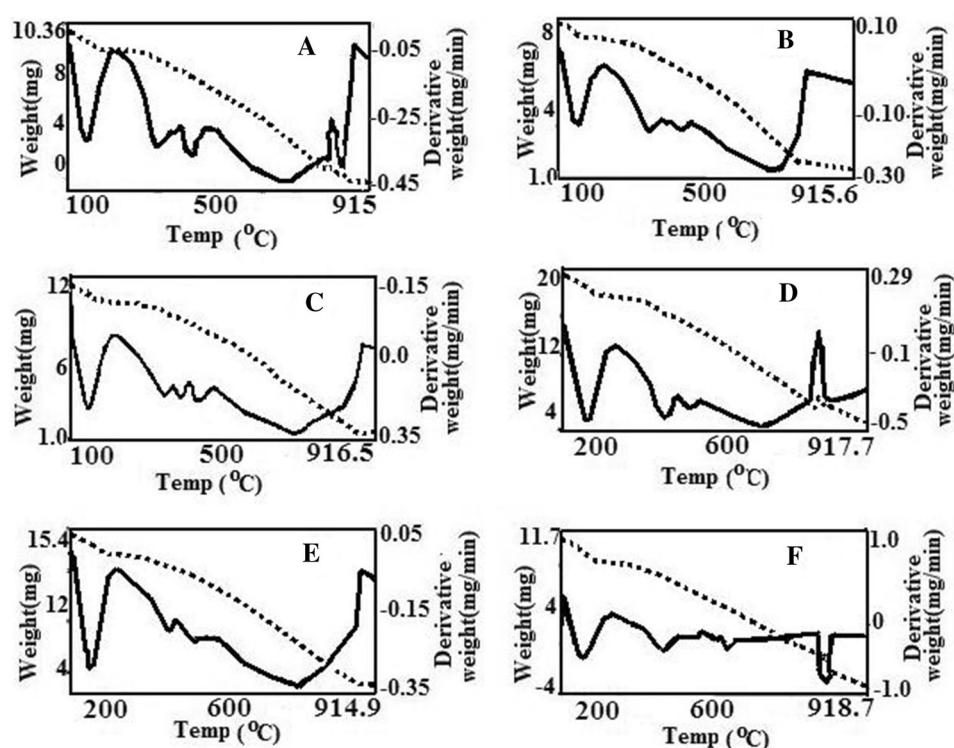


Table 2 TG/DTA data for the nanocomposites temperature (°C)

Sample	First step	Second step	Third step
PANiTNC1	91	300, 393	832
PANiTNC2	85	299	760
PANiTNC3	87	305	784
PANiTNC4	92	319	774
PANiTNC5	89	312	–
PANiTNC6	98	322	790

Effect of scan rate

The effect of varying scan rates from 50 to 500 mV/s was studied at the PANiTNC1-modified GCE in 1 M H₂SO₄. With increasing the scan rate, the peak current separation increased, and also the peak potential shifted slightly with the anodic peak to positive and the cathodic peak to negative potential directions. This is because the charging and discharging of the electroactive CP determines the rate. Figure 8d shows the plot of logarithm peak current against logarithm of scan rate, which clearly indicates that the current increases approximately in linear approach as described by the equation, $y = 0.596x - 1.1755$; $R^2 = 0.9933$ for the anodic peak currents. This also shows good adherence of the composite onto the electrode

surface. The experimental slope value of 0.596 was found to be greater than the theoretical value of 0.5, which suggests that process is purely adsorption controlled.

Chronoamperometry studies

Chronoamperometric studies were performed on PANiTNC1-modified GCE with normal, N₂- and O₂-saturated buffer with an initial and final potential of –1.6–1.4 V versus SCE. Figure 9 shows that the PANiTNC1-modified GCE current was high in O₂-saturated buffer which is however less in the case of normal and N₂-saturated buffer. Thus, the O₂ gas interfered with the nanocomposite layer and enhanced the current.

From the slope value obtained by the plot of I versus $t^{-1/2}$, the diffusion coefficient values of PANiTNC1-modified GCE were determined by Cottrell equation. The calculated surface coverage (N_p) and diffusion coefficient (D) values are tabulated in Table 4. The D and N_p values are very higher for O₂-saturated buffer in comparison to those for the normal and N₂-saturated buffer.

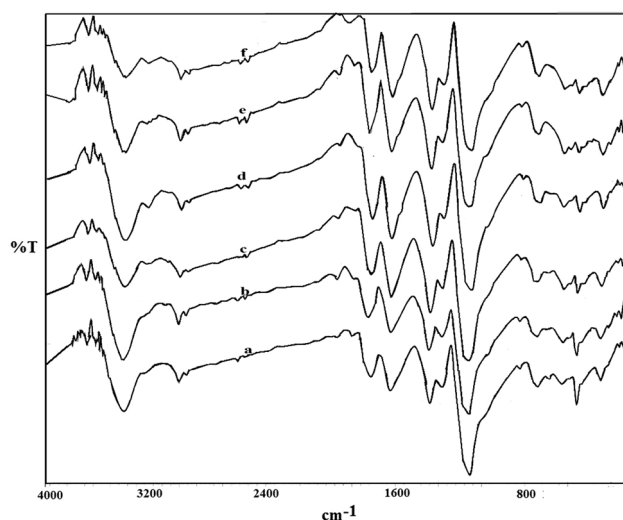
Chronocoulometry studies

Chronocoulometric behaviour of PANiTNC1-modified GCE was investigated in the absence and presence of N₂-

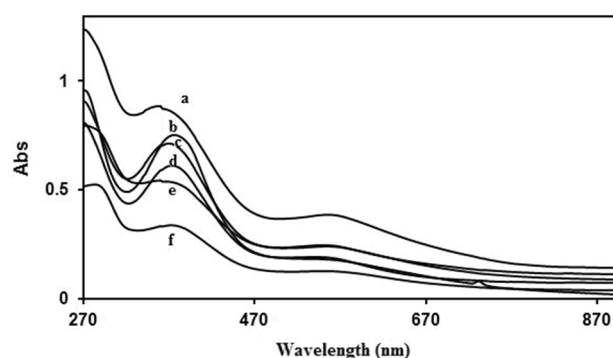


Table 3 Characteristic frequencies of PAniTNC1–PAniTNC6

PAniTNC1	PAniTNC2	PAniTNC3	PAniTNC4	PAniTNC5	PAniTNC6
3431	3433	3446	3449	3455	3456
2852	2852	2854	2854	2923	2923
1577	1579	1577	1577	1581	1577
1481	1477	1481	1485	1481	1481
1298	1298	1298	1298	1299	1301
1242	1242	1245	1245	1245	1244
1114	1116	1112	1120	1122	1126
798	800	802	800	804	802
721	723	724	724	726	725
505	505	505	505	505	505

**Fig. 6** FTIR spectra of PAniTNCs (*a* PAniTNC1, *b* PAniTNC2, *c* PAniTNC3, *d* PAniTNC4, *e* PAniTNC5 and *f* PAniTNC6)

and O₂-saturated buffer by performing a double potential step technique with initial and final potential of −1.4–1.2 V versus SCE (Fig. 10). PAniTNC1-modified GCE immersed in 1.0 M H₂SO₄ at 25 ± 2 °C for 2 s shows a large enhancement in charge in O₂-saturated buffer when compared to N₂ and normal buffer, indicating that the nitrogen and oxygen buffer interferes with the nanocomposite layer, thereby increasing the charge. It has been previously confirmed by cyclic voltammetry and chronoamperometric techniques. The Anson plot shows linear dependency of *Q* upon *t*^{1/2} indicating that the process involved adsorption controlled for normal condition as well as O₂- and N₂-saturated buffer solution.

**Fig. 7** UV–Vis spectra of PAniTNCs (*a* PAniTNC1, *b* PAniTNC2, *c* PAniTNC3, *d* PAniTNC4, *e* PAniTNC5 and *f* PAniTNC6)

Effect of pH in the presence of methanol

PAniTNC1-modified GCE shows enhanced electrocatalytic activity in 1.0 M H₂SO₄ (pH1) which slightly increases in the case of O₂-saturated buffer. For this reason, linear sweep voltammetry has been recorded in 1.0 M H₂SO₄ and 1.0 M MeOH (acidic methanol), on PAniTNC1-modified GCE. It was carried out in normal as well as O₂- and N₂-saturated in methanol at 50 mV/s shown in Fig. 11. The LSV shows a broad reduction peak from +698.6 to +303.8 mV for normal and N₂ atmospheric condition and two reduction peaks at +522.9 and +208.8 mV for O₂ medium. PAniTNC1-modified GCE shows a larger enhancement in the cathodic peak current and potential shifted towards the negative side (lower potential) under O₂ condition, suggesting good electrocatalytic behaviour. The results clearly show that PAniTNC1-modified GCE enhances electrocatalytic reduction (ORR activity) behaviour towards methanol under O₂-saturated acidic metha-



Fig. 8 **a** Plot of concentration versus current (1 PANiTNC1, 2 PANiTNC2, 3 PANiTNC3, 4 PANiTNC4, 5 PANiTNC5 and 6 PANiTNC6). **b** Plot of pH versus current. **c** Cyclic voltammetric responses obtained for PANiTNC1-modified GCE in the presence of pH 1.0 with a scan rate of 50 mV/s. **d** Plots of logarithm of scan rate versus logarithm peak current

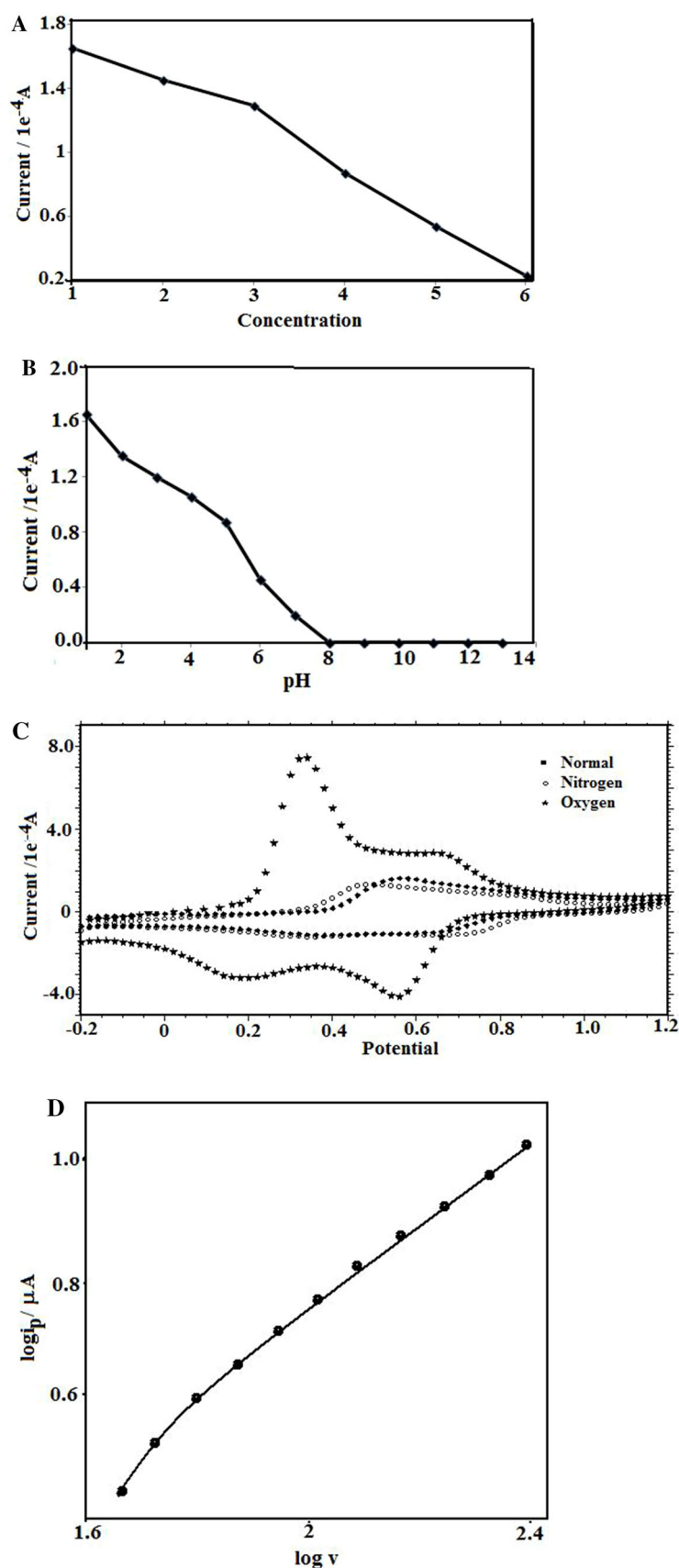


Fig. 9 Chronoamperogram of PANiTNC1 GCE at $25 \pm 2^\circ\text{C}$ with 250 ms

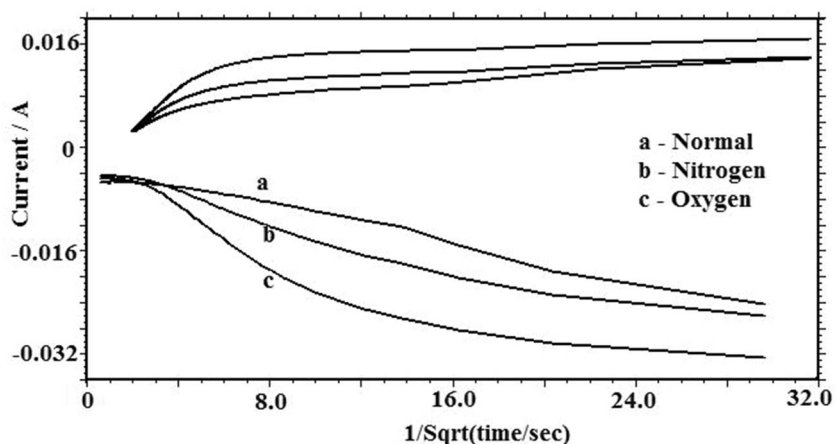
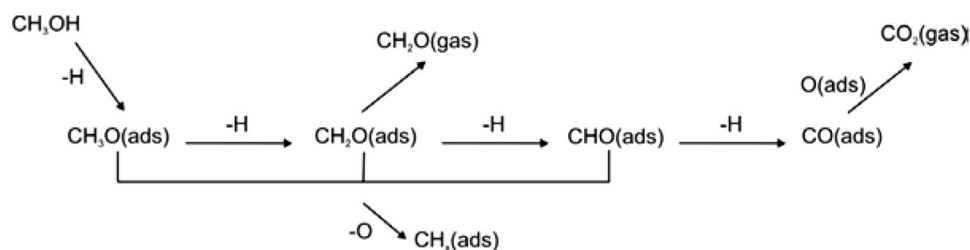


Table 4 Surface coverage (N_p) and diffusion coefficient of responses obtained for PANiTNC1-modified GCE

PAniTNC1	N_p (mol/cm ²)	D (cm ² /s)
Normal	4.0929×10^{-4}	2.9635×10^{-6}
N ₂	5.5122×10^{-4}	6.5829×10^{-6}
O ₂	1.208×10^{-3}	1.68913×10^{-5}

nol compared to that in pH 1.0 solution. The methanol reduction mechanism is given below.



Electrochemical impedance spectroscopy (EIS) studies

Electrochemical impedance measurements were carried over a frequency range from 1000 to 0.01 Hz in open circuit potential. Figure 12 shows the electrochemical impedance spectra (Nyquist plots, Z' vs. Z'') for PANiTNC1–PANiTNC6-modified GCE in 1.0 M H₂SO₄. The C_{dl} values of PANiTNC1–PANiTNC6 are given as follows: 3.442×10^{-8} , 8.48×10^{-8} , 2.69×10^{-9} , 4.115×10^{-9} , 4.6509×10^{-9} and 5.09×10^{-9} F/cm². The semicircle at high frequencies was characteristic of the charge-transfer process and the diameter of the semicircle

equals the charge-transfer resistance (R_{ct}) through PANi/TiO₂ composite. Smaller R_{ct} indicates a faster electron-transfer rate. The radii of the semicircle of PANiTNC1–PANiTNC6 (3.211×10^3 – 9.066×10^3) are found to increase with the increase in ceramic concentration which shows greater resistance to conduction with increase in ceramic concentration.

These values clearly indicate that a decrease in the capacitance values increases the resistance to Faraday current. Hence it can be concluded that the high conduc-

tivity of the nanocomposite with lower ceramic concentration (PANiTNC1) could enhance the electron-transfer rate. PANiTNC2–PANiTNC6 conductivity decreased gradually, while increasing the ceramic content. It is supported by UV and XRD studies.

Experimental

The aniline used was of analytical reagent (AR) grade from Sigma–Aldrich Co. Aniline was purified by distillation under vacuum before use. The PANiTNC nanocomposites were prepared by keeping the concentration of the monomer and oxidant constant and varying the wt% of ceramic



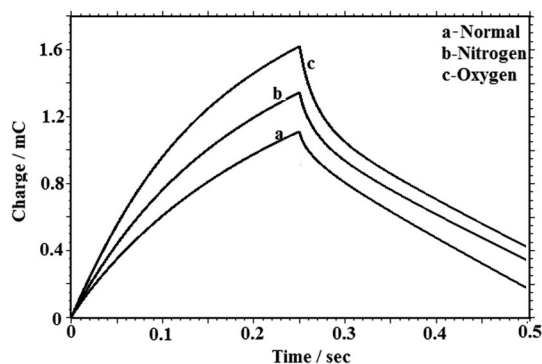


Fig. 10 Chronocouloumogram of PANiTNC1 GCE at 25 ± 2 °C with 250 ms

Fig. 11 Linear sweep voltammetric cathodic peak obtained for PANiTNC1-modified GCE in the presence of acidic methanol with a scan rate of 50 mV/s

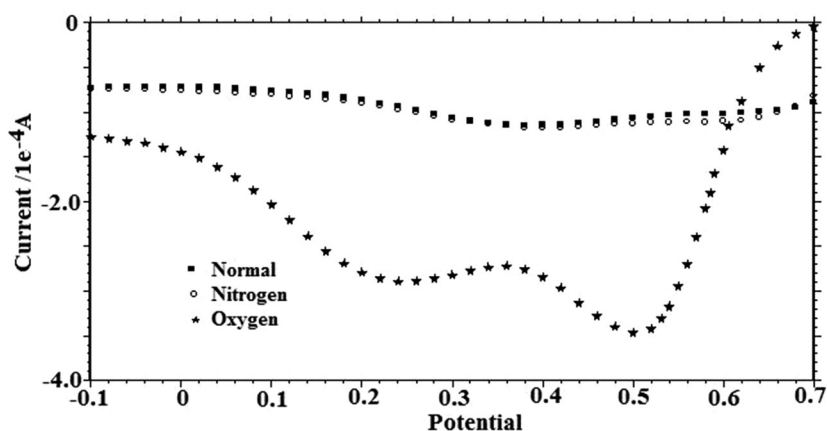


Fig. 12 Electrochemical impedance spectra of *a* PANiTNC1, *b* PANiTNC2, *c* PANiTNC3, *d* PANiTNC4, *e* PANiTNC5 and *f* PANiTNC6-modified GCE/TiO₂/PANI electrode in the presence of 1 M H₂SO₄ (pH 1.0)

from 0.25 to 1.5 g (0.25, 0.5, 0.75, 1.0, 1.25, 1.5 g). Bulk polymerization was carried out by mixing various weight percents of ceramic into a mixture of 9 ml of 0.5 M aniline and 6 ml of 1 M hydrochloric acid and made up to 200 ml using conductivity water. To this the oxidant viz., potassium persulphate was added with constant stirring. The nanocomposite temperature ≈ 0 °C was maintained for 24 h to complete the reaction. Then the precipitate was filtered and washed successively using distilled water until the wash solution turned colourless. The nanocomposite was dried in room temperature. The CP nanocomposites synthesized were designated as PANiTNC1 to PANiTNC6.

Conclusions

The various amount of TiO₂ used polyaniline–TiO₂ nanocomposites was synthesized successfully. The better results were obtained for lower concentration TiO₂ used

nanocomposites. Synthesised nanocomposites were analysed through UV-Vis and FTIR spectra. PANi–TiO₂ nanocomposites are in nanoscale, which is confirmed by XRD and TEM. SEM images showed that sponge-like morphology gradually changed into granular flower-like morphology. Ceramic content ratio was proved by EDX studies. Thermal stability of the PANiTNC was studied by TGA and DSC analyses. In electrochemical studies, PANiTNC1-modified GCE shows electrocatalytic redox behaviour under O₂-saturated buffer at pH 1.0. Especially, the reduction behaviour is enhanced for acidic methanol solution when subject to O₂-saturated buffer than N₂-saturated buffer and normal condition. Voltammograms show that redox reaction of PANiTNC1 is adsorption controlled. The diffusion coefficient value is high in the case of O₂-saturated buffer. This has also been depicted by chronoamperometric and chronocoulometric studies. EIS study also showed that PANiTNC1 exhibits good conductivity.



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Authors' contributions PR carried out synthesis and all experimental works. CV monitored all the works. Authors have read and approved the final manuscript.

Conflict of interest The authors declare that they have no competing interests.

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